

Delocalization Error: The Greatest Outstanding Challenge in Density-Functional Theory

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Everyday, density-functional theory (DFT) is routinely applied to computational modeling of molecules and materials with the expectation of high accuracy. However, in certain situations, popular density-functional approximations (DFAs) have the potential to give substantial quantitative, and even qualitative, errors. The most common class of error is delocalization error, which is an overarching term that also encompasses the one-electron self-interaction error. In our opinion, its resolution remains the greatest outstanding challenge in DFT development. In this paper, we review the history of delocalization error and provide several complementary conceptual pictures for its interpretation, along with illustrative examples of its various manifestations. Approaches to reduce delocalization error are discussed, as is its interplay with other shortcomings of popular DFAs, including treatment of non-bonded repulsion and neglect of London dispersion.

1. INTRODUCTION

Kohn-Sham density-functional theory (KS-DFT)¹ is now the workhorse of computational chemistry, solid-state physics, and materials science. Papers proposing some of the most popular functionals constitute 12 of the top 100 most-cited scientific papers, across all fields.² For the majority of chemical systems, DFT methods provide an excellent balance between computational cost and accuracy. They are highly successful in modeling reaction thermochemistry and kinetics, proposing reaction mechanisms, investigating dynamics, modeling solid-state structure, and predicting phase transitions, to list a few examples. However, for certain classes of systems, popular KS-DFT methods have the potential to fail dramatically. One such class of system is those with multi-reference character,^{3–9} meaning that the wavefunction cannot be well represented by a single electronic configuration (i.e. Slater determinant). Development of DFT methods for multi-reference systems has been the subject of much recent research,^{10–24} although they are relatively rare in ground-state chemistry.

More commonly, errors are seen for systems with highly delocalized electron densities. The most notable examples are H_2^+ , with its bond length stretched far beyond equilibrium,^{4,25–28} and analogous stretched ions (He_2^+ , Li_2^+ , Ne_2^+ , etc.). However, similar errors have been identified for many other systems, including charge-transfer complexes,^{29–32} transition states of radical reactions,^{33–38} band gaps of semi-conductors,^{39–44} polarizabilities of long-chain molecules,^{45–48} systems with extended conjugation,^{49–58} halogen and chalcogen bonds,^{59–61} and organic acid/base co-crystals.⁶² These are not the result of separate errors, but rather represent

many facets of one single error in common density functionals. This error was originally termed self-interaction error (SIE)⁶³ and discussed in the context of a single electron. However, its definition was later expanded^{64–66} and the more general case is known as delocalization error.⁶⁷ The latter is the more inclusive term, since delocalization error may still affect results that have been corrected for (one-electron) SIE. While many methods have been proposed over the past four decades to limit delocalization error, no one approach is a panacea and research in this area is ongoing.^{68–75} Resolving delocalization error almost certainly remains the greatest outstanding challenge in DFT.

In this work, we will review the history of delocalization error, describing three different perspectives on its theoretical underpinnings. We will discuss how this error can be analyzed in terms of density and energy contributions. Further, we will consider the relation between delocalization error and dispersion corrections, and emphasize the importance of obtaining accurate predictions for the right physical reasons. Finally, we will provide a brief survey of some classes of functionals undergoing current development that offer promise in reducing delocalization error.

2. THEORETICAL BACKGROUND

2.1. Overview of DFT

In KS-DFT, the energy of a chemical system can be obtained from the electron density and occupied KS orbitals. The total energy is

$$E = T_0 + V_{\text{nuc}} + J_{\text{ee}} + E_{\text{XC}}. \quad (1)$$

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T_0 is the non-interacting kinetic energy,

$$T_0 = -\frac{1}{2} \sum_{\sigma} \sum_i \int \psi_{i\sigma}(\mathbf{r}) \nabla^2 \psi_{i\sigma}(\mathbf{r}) d\mathbf{r}, \quad (2)$$

where σ denotes electron spin and the ψ 's are the occupied KS orbitals, assumed to be real. V_{nuc} is the potential energy from electron-nuclear interactions,

$$V_{\text{nuc}} = \sum_{\sigma} \int \rho_{\sigma}(\mathbf{r}) V_{\text{ext}} d\mathbf{r}, \quad (3)$$

where V_{ext} is the external potential from the nuclei and ρ_{σ} is the σ -spin electron density, which can be determined from the real, occupied KS orbitals as

$$\rho_{\sigma}(\mathbf{r}) = \sum_i \psi_i^2(\mathbf{r}). \quad (4)$$

J_{ee} is the classical Coulomb energy for electron-electron interactions,

$$J_{\text{ee}} = \frac{1}{2} \iint \frac{1}{r_{12}} \rho(\mathbf{r}_1) \rho(\mathbf{r}_2) d\mathbf{r}_1 d\mathbf{r}_2, \quad (5)$$

where r_{12} is the electron-electron distance. Finally, E_{XC} is the exchange-correlation energy, which is usually written as a functional of the density and its derivatives. This latter term may be further divided into separate exchange (E_{X}) and correlation (E_{C}) energies.

KS-DFT is exact, in principle, provided that one can employ the exact exchange-correlation functional. However, this functional is unknown and there is no systematic route to obtain it.^{1,76} In practise, various approximations to E_{XC} are classified by their dependence on the density. The simplest density-functional approximation (DFA) is the local spin-density approximation (LSDA),^{1,63,77-79} where E_{XC} depends only on the density at each point on a real-space grid.⁸⁰ The next most sophisticated DFAs are generalized gradient approximations (GGAs), where E_{XC} depends on the density and its gradient at each point. Many different GGAs have been proposed (B86a,⁸¹ B86b,⁸² PW86,⁸³ B88,⁸⁴ PW91,⁸⁵ PBE,⁸⁶ PBEsol,⁸⁷ etc.) and the choice of GGA is particularly important for describing non-bonded repulsion.⁸⁸⁻⁹⁶ GGAs are the standard models employed in solid-state DFT using planewave/pseudopotential codes.

Beyond GGAs, meta-GGAs also include dependence on the kinetic-energy density (τ), and occasionally on the Laplacian of the density as well.⁹⁷⁻¹⁰⁷ Functions of the kinetic-energy density can be used to describe electron delocalization,^{108,109} so meta-GGAs aim to include some correction for self-interaction at close to GGA cost. There are two general philosophies behind the development of meta-GGAs. Many meta-GGAs, most notably M06L¹⁰³ and related Minnesota functionals, use power-series expansions of the exchange and correlation energies with empirical parameters fit to thermochemical reference data. Alternatively, others are designed to obey cer-

tain physical constraints, including the popular SCAN¹⁰⁶ functional and its recent revision, r²SCAN¹⁰⁷, with improved numerical stability. While both types tend to perform relatively well for thermochemistry benchmarks such as GMTKN55,^{110,111} use of the more complicated functional forms of meta-GGAs does not necessarily lead to improved accuracy.^{112,113} As we will see, the LSDA, GGAs, and meta-GGAs all have significant inherent delocalization error,^{28,110,114} albeit to differing extents.¹¹⁵⁻¹¹⁸

Next is the class of hybrid functionals,¹¹⁹ which include the exact exchange energy from Hartree-Fock theory. Global hybrids involve a constant, position-independent mixing of density-functional and exact exchange of the form

$$E_{\text{X}}^{\text{hybrid}} = a_{\text{X}} E_{\text{X}}^{\text{HF}} + (1 - a_{\text{X}}) E_{\text{X}}^{\text{DFA}}, \quad (6)$$

where $0 < a_{\text{X}} < 1$ is the mixing parameter. These are the most commonly used methods in molecular DFT. Hybrids designed for general thermochemistry, such as B3LYP,¹²⁰⁻¹²² B97,^{123,124} and PBE0,^{86,125} typically use ca. 20-25% exact exchange. Hybrids designed to reduce delocalization error for chemical kinetics, such as MPW1K¹²⁶ and M06-2X,¹²⁷ use larger fractions, near 50% exact exchange.

Building on global hybrid functionals are range-separated hybrids,¹²⁸⁻¹³⁷ where the degree of exact-exchange mixing is dependent on the inter-electron distance (r_{12}). This is achieved by separating the exchange terms into short-range and long-range components via an error function,

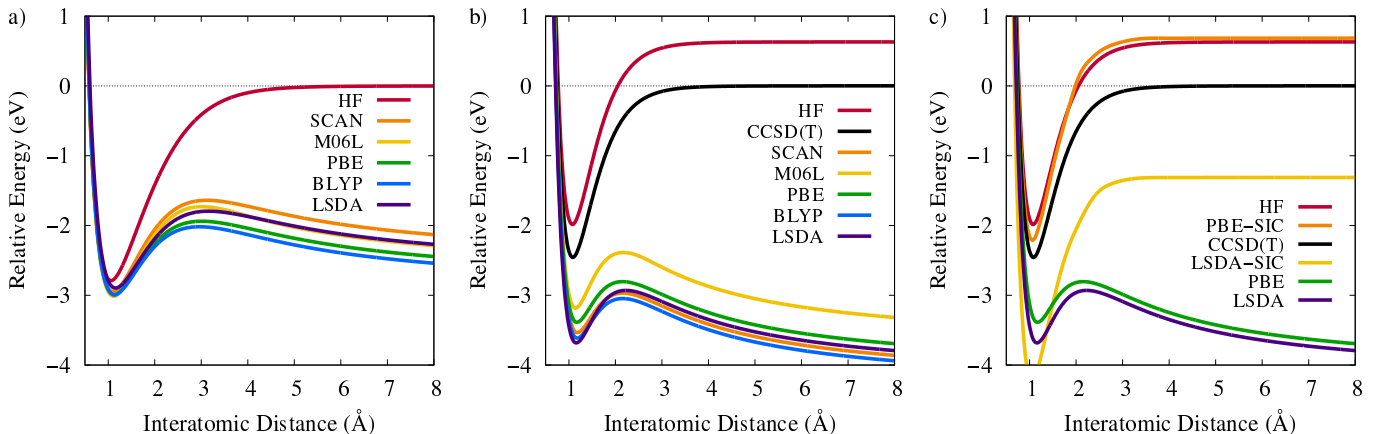
$$\frac{1}{r_{12}} = \underbrace{\frac{1 - \text{erf}(\omega r_{12})}{r_{12}}}_{\text{short range}} + \underbrace{\frac{\text{erf}(\omega r_{12})}{r_{12}}}_{\text{long range}}, \quad (7)$$

where ω is the range-separation parameter. The exchange-correlation energy is evaluated for the two components separately, using different methods, and the results then recombined. In periodic systems, functionals with full GGA exchange at long range, such as HSE,^{136,137} are used to reduce the computational cost associated with hybrid DFT. In molecular systems, functionals with full HF exchange at long range, such as LC-BLYP,¹³⁰ LC- ω PBE,^{133,134} and ω B97X,¹³⁵ are used to minimize delocalization error. While there are additional classes of functionals that depend on the exact exchange-energy density, or the unoccupied (virtual) KS orbitals, these will not be discussed until Section 5.

2.2. Self-Interaction Error

Self interaction arises from the Coulomb energy terms that correspond to interaction of an electron with itself. For a one-electron system, the Coulomb energy given by Eqn 5 is non-zero, which is not physical. However, in Hartree-Fock theory, the non-zero Coulomb energy for

FIG. 1. Potential energy curves for (a) H_2^+ and (b) He_2^+ obtained with selected methods and the aug-cc-pVTZ basis set using Gaussian 16¹³⁸ (psi4¹³⁹ for the SCAN calculations). Also shown in (c) are LSDA and PBE potential energy curves for He_2^+ with the Perdew-Zunger self-interaction correction⁶³ (SIC) obtained using the PyFLOSIC code.¹⁴⁰



interaction of an electron with itself is offset by a corresponding exchange-energy term that is equal in magnitude, but opposite in sign. The exact (Hartree-Fock) exchange energy is

$$E_X = \frac{1}{2} \sum_{\sigma} \iint \frac{1}{r_{12}} \psi_{i\sigma}(\mathbf{r}_1) \psi_{j\sigma}(\mathbf{r}_2) \psi_{j\sigma}(\mathbf{r}_1) \psi_{i\sigma}(\mathbf{r}_2) d\mathbf{r}_1 d\mathbf{r}_2, \quad (8)$$

where σ indicates electron spin and the ψ 's are the occupied KS orbitals, assumed to be real. Thus, the HF exchange energy is also non-zero for a single electron.

For a one-electron system, with a single occupied orbital, the requirement for vanishing self-interaction energy is⁶³

$$J_{ee} + E_X = 0. \quad (9)$$

This condition is satisfied by the exact (HF) exchange energy due to the offsetting Coulomb and exchange integrals. However, for DFAs, such as the LSDA or GGAs, the exchange energy does not offset the self Coulomb energy. Indeed, the DFA exchange energy can be very much larger in magnitude than the exact exchange energy in some systems, resulting in over-stabilization.

The classic example of this is the dissociation limit of the H_2^+ potential energy curve, which is described exactly by HF theory. However, this dissociation limit is predicted to be far too low in energy by most DFAs,^{4,25,26,28} as shown in Figure 1(a) for the LSDA,⁶³ and for BLYP^{84,121} and PBE,⁸⁶ which are representative GGA functionals. Results are also shown for two meta-GGAs: M06L¹⁰³ and SCAN¹⁰⁶. The dissociation limit obtained with each DFA is far below the exact result from HF theory. Figure 1(b) shows analogous results for He_2^+ . Here, HF is not exact since He_2^+ is not a one-electron system and, due to neglect of electron correlation effects, it gives a dissociation limit above the CCSD(T) reference. All the DFAs again fail catastrophically, predicting He_2^+ with a highly stretched bond length to be far lower in

energy than the separated He atom and He^+ ion. These results are not unique to the specific functionals chosen; all GGAs and meta-GGAs show similar errors.²⁸

Self-interaction error can be reduced through use of hybrid functionals, since they include some fraction of exact exchange. However the optimum exchange-mixing fraction is clearly system dependent. For a uniform electron gas, the LSDA is exact and there is no self-interaction error. Bulk metals are the systems that most closely resemble a uniform electron gas, and are typically very well treated by the LSDA and GGA functionals. Conversely, for one-electron systems, full exact exchange is needed to eliminate self-interaction error. Most many-electron systems fall somewhere between these two extremes, where some optimal mixing of DFA and HF exchange will minimize SIE. As a result, SIE is identified in practise in cases where the LSDA or GGAs drastically over-stabilize a particular system and the error decreases with increasing exact-exchange mixing, usually up to ca. 50% HF exchange for a global hybrid or 100% long-range HF exchange for a range-separated hybrid.

2.3. Self-Interaction Corrections

To improve performance of the LSDA (or GGAs), Perdew and Zunger⁶³ developed a one-electron self-interaction correction (SIC). This correction uses single-orbital spin densities,

$$\rho_{i\sigma}(\mathbf{r}) = \psi_{i\sigma}^2, \quad (10)$$

such that the total density is a sum of all the one-electron spin densities. The self-interaction-corrected exchange-correlation energy is then

$$E_{XC}^{\text{SIC}} = E_{XC}(\rho_{\alpha}, \rho_{\beta}) - \sum_{i\sigma} [J_{ee}(\rho_{i\sigma}) - E_{XC}(\rho_{i\sigma})], \quad (11)$$

where ρ_α and ρ_β are the α - and β -spin indexed electron densities. The term in square brackets resembles the left-hand side of Eqn. 9 and is the self-interaction correction for orbital i with spin σ . This term would be identically zero for either HF or the exact exchange-correlation functional. An analogous SIC can be formulated for the exchange-correlation potential, which allows solution for a set of corrected KS orbitals and energy eigenvalues.⁶³ The correction to the eigenvalues allows the SIC method to improve DFA predictions of band gaps.^{141–145}

For the H_2^+ example, any SIC corrected functional will necessarily give the exact (Hartree-Fock) result since Eqn. 11 is designed to recover the HF energy for one-electron systems. However, Eqn. 11 provides only an approximate correction for many-electron systems. As shown in Figure 1(c), the SIC corrects the shape of the He_2^+ PES when applied to both PBE and the LSDA, although it does not recover the exact dissociation limit corresponding to the sum of the separated He and He^+ energies.

Unfortunately, the SIC tends to worsen predicted properties of molecules exhibiting minimal SIE,^{146–150} so a scaled SIC of the form

$$E_{\text{XC}}^{\text{SIC}} = E_{\text{XC}}(\rho_\alpha, \rho_\beta) - \sum_{i\sigma} X_{i\sigma} \left[J_{ee}(\rho_{i\sigma}) - E_{\text{XC}}(\rho_{i\sigma}) \right] \quad (12)$$

was introduced.¹⁵¹ Here, $X_{i\sigma}$ is a scaling factor,

$$X_{i\sigma} = \int z_\sigma^k \rho_{i\sigma}(\mathbf{r}) d\mathbf{r}, \quad (13)$$

with

$$z_\sigma = \frac{\tau_\sigma^{\text{W}}(\mathbf{r})}{\tau_\sigma(\mathbf{r})} \quad (14)$$

such that $0 < X_{i\sigma} \leq 1$. $\tau_\sigma = \frac{1}{2} \sum_i |\nabla \psi_{i\sigma}|^2$ is the σ -spin kinetic-energy density, τ_σ^{W} is the von Weizsäcker kinetic-energy density,

$$\tau_\sigma^{\text{W}}(\mathbf{r}) = \frac{|\nabla \rho_\sigma(\mathbf{r})|^2}{8\rho_\sigma(\mathbf{r})}, \quad (15)$$

and $k \geq 0$ is a scaling exponent. This form recovers the original PW SIC for $k = 0$ and the uncorrected DFA result in the limit of $k \rightarrow \infty$.¹⁵¹ In practise, k values of 1 or 2 were chosen to provide improved performance for molecular properties.^{151,152}

This scaling does not affect the H_2^+ PES as $\tau = \tau^{\text{W}}$ for all one-electron systems. The scaling also has only a minor effect on the computed PES for He_2^+ .¹⁵² However, it was found that reduction of the SIC through this scaling substantially degrades performance for the PES of Ne_2^{+152} and for NaCl .⁶⁶ In the latter case, the LSDA and GGAs predict a dissociation limit with spurious fractional charges on both atoms. Ultimately, this leads to a similar situation seen with the amount of exact-exchange mixing in hybrid functionals, where no

single choice of %HF or k works well for all systems. Optimizing k for thermochemistry still results in significant SIE, while taking $k = 0$ to minimize SIE worsens molecular properties, hampering the general applicability of self-interaction corrections. One solution is to introduce a local, position-dependent scaling.^{153–155} In the resulting local self-interaction correction (LSIC) scheme, the corrected exchange-correlation energy is analogous to Eqn. 11, but with the J_{ee} term replaced by

$$U^{\text{LSIC}}(\rho_{i\sigma}) = \frac{1}{2} \iint f[z_\sigma(\mathbf{r}_1)] \frac{1}{r_{12}} \rho_{i\sigma}(\mathbf{r}_1) \rho_{i\sigma}(\mathbf{r}_2) d\mathbf{r}_1 d\mathbf{r}_2, \quad (16)$$

and the E_{XC} term replaced by

$$E_{\text{XC}}^{\text{LSIC}}(\rho_{i\sigma}) = \int f[z_\sigma(\mathbf{r})] \varepsilon_{\text{XC}}[\rho_{i\sigma}(\mathbf{r})] d\mathbf{r}, \quad (17)$$

where ε_{XC} is the exchange-correlation energy density. $f(z_\sigma)$ is some function with values between 0 and 1, with the simplest alternative being z_σ^k . It can also be shown that this local scaling can be written as a global scaling after a change of gauge.¹⁵⁶

The additional key drawback of the self-interaction correction is that it is orbital dependent and, hence, is not invariant with respect to unitary orbital transformations. This means that the value of the SIC will depend on the choice of orbitals. Using the KS orbitals is intuitive, but their highly non-local nature means that the SIC will approach zero in the bulk limit.^{63,157} This can be resolved by using localized orbitals, but there are many possible localization schemes and the magnitude of the SIC will depend on the particular one selected. Recent development work has focused on Fermi-Löwdin orbitals (FLO),^{140,157–159} which provide a convenient framework for general application of the SIC to chemical systems of any size.^{160–164} Ongoing development has involved orbital-dependent scaling,^{151,165} as well as use of FLOSIC with the density-consistent effective potential^{166,167} or the Krieger-Li-Iafrate (KLI)^{168,169} approximation to the optimized effective potential.

2.4. The Fractional-Charge Picture

An alternative, but complementary, view of SIE considers gradual ionization of ensembles of atoms or molecules. Removal (or addition) of one electron from an ensemble of symmetry-equivalent atoms will result in a fractional charge on each atom in the ensemble.^{64,67,115,170,171} Assuming the atoms are sufficiently far apart to be non-interacting, then the energy should be equivalent to that of an isolated ion and many neutral atoms. For our stretched H_2^+ example, each H atom will have a fractional charge of +0.5 and the energy should be equal to that of the separated H atom and H^+ ion, which is the correct dissociation limit. We can extend the number of atoms, n , in the ensemble and remove the appropriate number of electrons, $p < n$, to

obtain any fractional charge, $q = p/n$, on each H atom, giving $(\text{H}^{+q})_n$. The energy of such an ensemble should then be equal to the sum of the energies of $(n - p)$ H atoms and p H^+ ions. Similarly, if we add $p < n$ electrons, the fractional charge would be $q = -p/n$, resulting in an ensemble energy equal to $(n - p)$ H atoms and p H^- ions. More generally, it can be proven that the total energy for any atom should be piecewise linear in fractional charge, q , with derivative discontinuities occurring at the integer values.⁶⁴

In practise, fractional-charge calculations cannot be performed by most electronic-structure codes. However, such curves can be drawn by using a cubic spline of the form^{172,173}

$$E(x) = E_{N-1} + \Delta E(x + 1 - N) + (N - x)(x + 1 - N) \left[(\varepsilon_{N-1}^{\text{LUMO}} - \Delta E)(N - x) + (\Delta E - \varepsilon_N^{\text{HOMO}})(x + 1 - N) \right], \quad (18)$$

where x is the total number of electrons, such that $N - 1 \leq x \leq N$, and $\Delta E = E_N - E_{N-1}$. The energies of the integer-charged species (E_N and E_{N-1}) are computed by conventional electronic-structure calculations. The slope at $x = N - 1$ is the LUMO (lowest unoccupied molecular orbital) eigenvalue of the $(N - 1)$ -electron species, $\varepsilon_{N-1}^{\text{LUMO}}$, while the slope at $x = N$ is the HOMO (highest occupied molecular orbital) eigenvalue of the N -electron species, $\varepsilon_N^{\text{HOMO}}$, in accordance with Janak's theorem.¹⁷⁴

As an illustration, we show the energy of the Li atom as a function of varying number of electrons from Li^+ ($x = 2$) to Li^- ($x = 4$) in Figure 2 for several selected DFAs. Also shown is the deviation from the ideal, linear behaviour, obtained by omitting the first two terms from the right hand side of Eqn. 18. The largest deviations from linearity are seen for the GGA functional (PBE), which gives concave-up curves and over-stabilizes the fractionally charged species. The excess stabilization of $\text{Li}^{+0.5}$ implies that the dissociation limit of Li_2^+ will also be too low in energy, similar to what was seen in Figure 1 for H_2^+ and He_2^+ .

The fractional-charge interpretation of delocalization error is particularly convenient to explain the band-gap problem.^{41,42} This problem refers to the tendency of many DFAs, particularly the LSDA and GGAs, to underestimate the band gaps of semi-conductors,^{39,40} as illustrated in Figure 3 for bulk silicon. The molecular equivalent of the band gap is the HOMO-LUMO gap, which is the difference between the HOMO and LUMO eigenvalues. The orbital eigenvalues are the derivatives of the energy with respect to electron number,¹⁷⁴

$$\varepsilon^{\text{HOMO}} = \lim_{x \rightarrow N-} \frac{E(x) - E(N)}{x - N} \quad (19)$$

$$\varepsilon^{\text{LUMO}} = \lim_{x \rightarrow N+} \frac{E(x) - E(N)}{x - N}, \quad (20)$$

such that the HOMO energy is the derivative from below, the LUMO energy is the derivative from above, and there is a derivative discontinuity at the integer points.⁶⁴ Since plots of the LSDA and GGAs energies with respect to fractional charges are concave up, the HOMO is too high in energy (i.e. the slope is not steep enough) and the LUMO is too low in energy (i.e. the slope is too steep), as illustrated in Figure 4. This results in closing of the HOMO-LUMO gap in a finite molecular system with GGAs and, hence, the band gap in a bulk solid.

Returning to Figure 2, the deviations from linearity seen with the PBE GGA are reduced by inclusion of some fraction of exact (Hartree-Fock) exchange, as in PBE0 (25% HF exchange). Including large fractions of exact exchange, as in LC- ω PBE (100% HF exchange for large r_{12}), can reverse the sign of the error and give concave-down curves, under-stabilizing the fractionally charged species. This can be referred to as a ‘‘localization’’ error, in contrast to delocalization error. Localization errors seen with HF theory itself stem from neglect of electron correlation rather than an inexact form of the exchange functional. This neglect of correlation results in HF's vast overestimation of the band gap seen in Figure 3, and a similar localization error is seen with HF for the dissociation limit of He_2^+ in Figure 1. MP2 and other correlated wavefunction theories are effectively free of delocalization or localization error and their energies vary (nearly) linearly with fractional charge.¹⁷⁷ In range-separated hybrid functionals, long-range, non-local correlation is required to be compatible with the full long-range exact exchange, and its absence leads to a localization error, similar to that seen with HF.

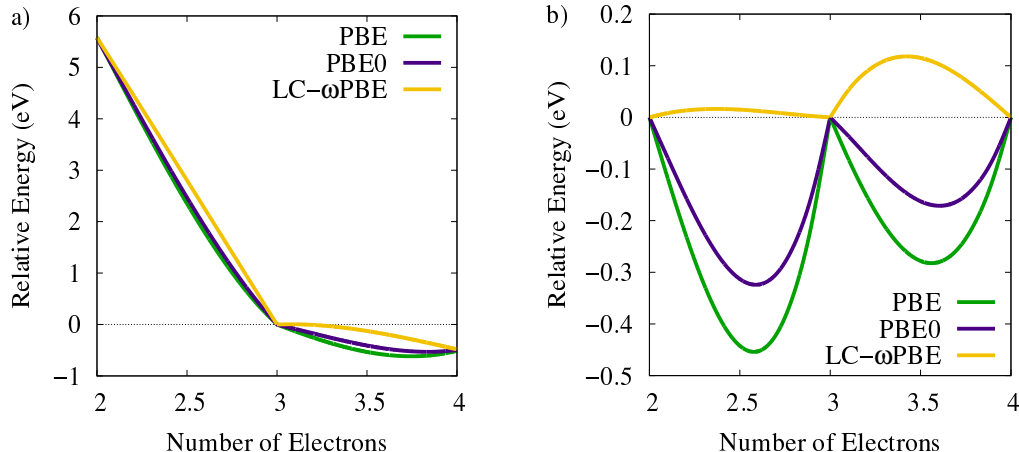
HF theory and some density functionals designed to reduce delocalization error^{63,178–180} give energies that vary linearly with fractional charge for one-electron systems by construction. However, such methods display non-linear behaviour for many-electron systems and hence still have inherent errors^{65,66,152,181} that affect predictions of thermochemistry, reaction barrier heights, band gaps, polarizabilities, charge transfer, and other properties. This lead to the distinction between one-electron and many-electron SIE,^{65,66,152,181} while delocalization error is often used as a more general, aggregate term. A DFA is defined to be ‘‘ N -electron self-interaction free’’ if the energy varies linearly with fractional electron number, x , such that

$$E(x) = E_{N-1}(N - x) + E_N(x + 1 - N) \quad (21)$$

for $N - 1 \leq x \leq N$. Functionals that are free of 1-electron SIE will give the correct dissociation limit for H_2^+ and tend to perform well for H-atom transfer barriers.^{35,36,178,182,183} However, functionals capable of correcting for many-electron SIE (i.e. for some general $N \geq 2$) are required more frequently and this remains an open area for functional development.

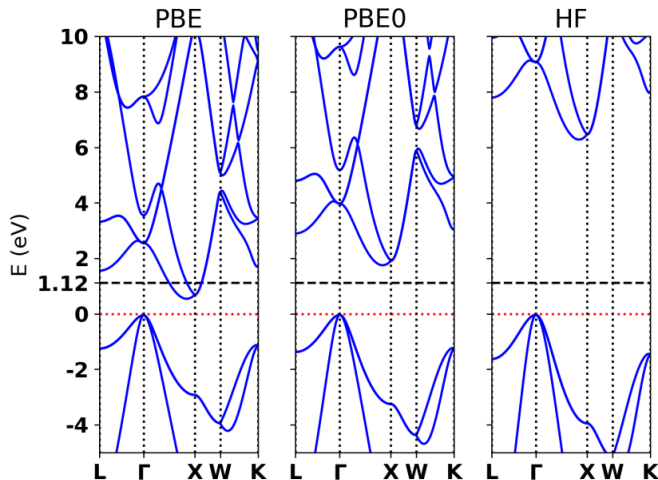
Piecewise linearity of the total energies is now a common test to assess the performance of new functionals.

FIG. 2. Relative energies for Li as a function of electron number; (a) total energies and (b) deviation from ideal, linear behaviour. Calculations were performed using Gaussian 16 and the aug-cc-pVTZ basis set.¹³⁸



Notably, range-separated hybrids are typically capable of satisfying Eqn. 21 for a given molecule with a certain choice of the range-separation parameter (ω in Eqn. 7). System-specific ω -tuning, such as to enforce Koopmans' theorem that the ionization potential equal the HOMO eigenvalue, can be used to obtain improved excitation energies and other properties.^{184–196} However, as the particular ω value required to minimize N -electron SIE is system dependent,^{173,197–200} the piecewise-linear energy constraint cannot be satisfied simultaneously for a diverse set of molecules. The recent localized orbital scaling correction (LOSC)^{201–205} was developed to enforce

FIG. 3. Dependence of the computed band structure of bulk silicon on the extent of exact-exchange mixing. Calculations were performed using all-electron electronic structure code FHI-aims.¹⁷⁵ All calculations use the numerical atom-centred orbital basis sets with the FHIaims 'light' settings and a $12 \times 12 \times 12$ k-space grid. A lattice constant of 5.416 Å was used. The zero of energy is taken as the top of the valence band in all cases and experimental band gap of 1.12 eV¹⁷⁶ is shown for comparison.



the piecewise-linearity constraint and shows good performance for many common systems that showcase delocalization error.

2.5. The Exchange-Correlation Hole Picture

A final way to understand SIE is through visualization of the exchange-correlation hole.^{4,206,207} Given an electron at a fixed reference point, the hole measures the depletion in probability of finding a second electron a certain distance away from that point. HF theory gives an exchange hole that is delocalized over all atoms in a system, but correlation usually serves to localize the full exchange-correlation hole to within regions of roughly atomic size.^{4,208} Conversely, most DFAs, and particularly the LSDA and GGAs, already model exchange holes as being fairly localized about each reference point.^{77,85,97} This is typically advantageous as the model holes implicitly build in non-dynamical correlation effects. However, in situations with significant SIE, non-dynamical correlation is insufficient to localize the hole and the exact exchange-correlation hole remains delocalized, in a similar way to the HF hole. In such cases, the assumptions made in construction of model DFA exchange holes are no longer valid.

As an example, we contrast the hydrogen atom, where DFAs give effectively zero SIE, with the H_2^+ dissociation limit. From the discussion in Section 2.4, the dissociation limit of H_2^+ is equivalent to two $\text{H}^{+0.5}$ cations. For simplicity, the electron density of $\text{H}^{+0.5}$ is approximated as half the H-atom density. Figure 5(a,b) illustrates these two density distributions.

Since there is no electron correlation in either H or $\text{H}^{+0.5}$, we can focus entirely on the exchange hole. We use the Becke-Roussel⁹⁷ (BR) meta-GGA for this demonstration since it is conveniently based on a real-space representation of the exchange hole as a simple exponential function displaced from the reference point. It is also

FIG. 4. Illustration of how fractional-charge errors with common DFAs lead to band-gap closing, for the example of the Li atom.

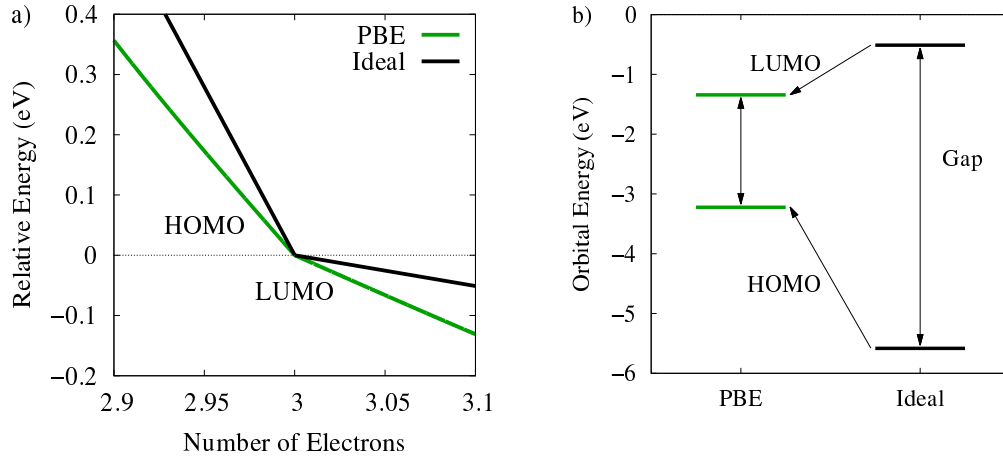
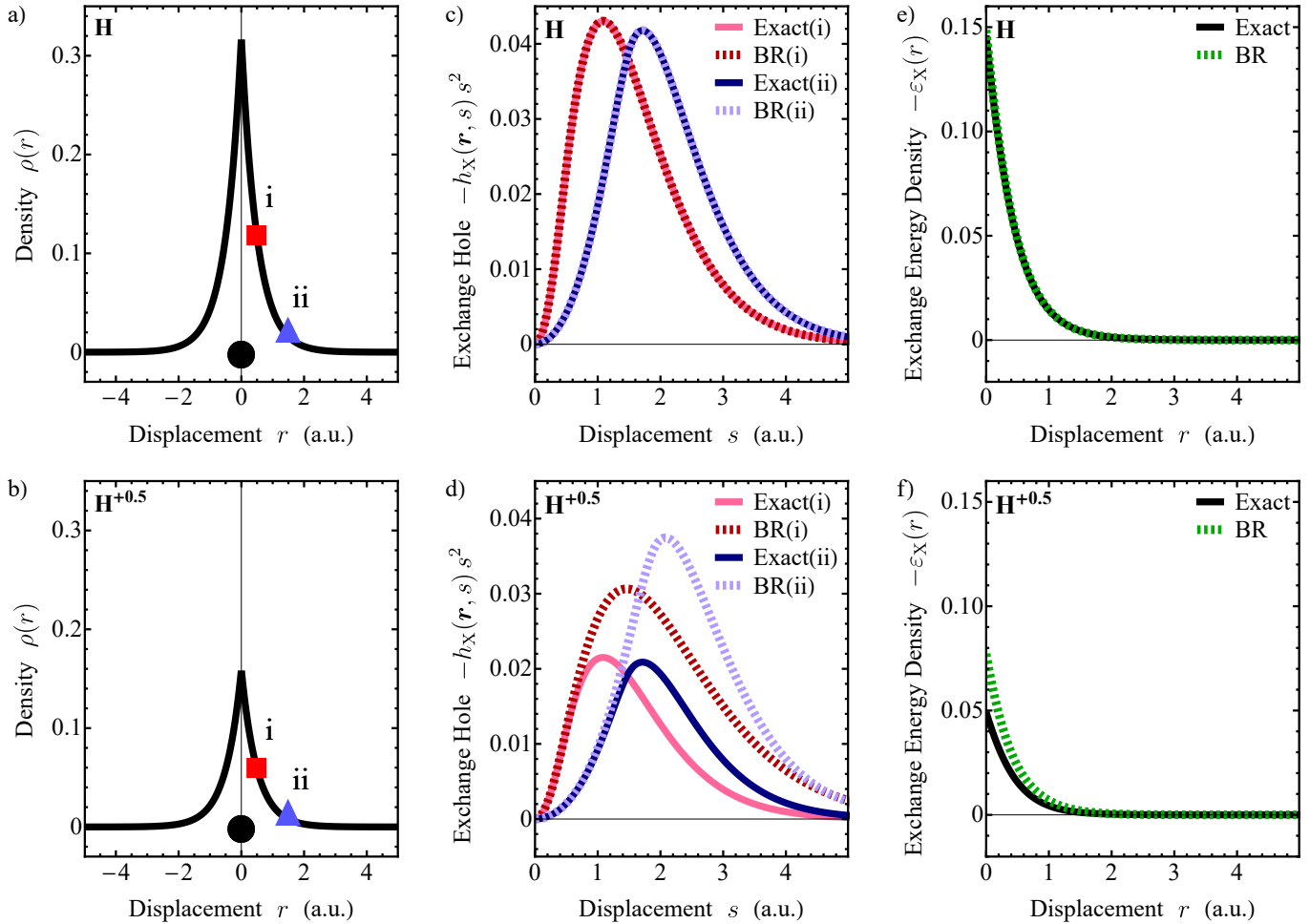


FIG. 5. Plots of the total electron density (a,b) and the normalization integrand for spherical averages of BR and exact exchange holes (c,d) about the two shown reference points for H (top row) and $H^{+0.5}$ (bottom row). Also shown are the exchange-energy densities for both species (e,f). All quantities are given in Hartree atomic units (a.u.).



constrained to recover the density and curvature of the exact exchange hole at the reference point. The DFT exchange energy is uniquely determined by the exchange hole via

$$E_X = \frac{1}{2} \sum_{\sigma} \iint \frac{\rho_{\sigma}(\mathbf{r}_1)}{r_{12}} h_{X\sigma}(\mathbf{r}_1, \mathbf{r}_2) d\mathbf{r}_1 d\mathbf{r}_2. \quad (22)$$

It can also be shown that the exchange energy is dependent only on the spherical average of the hole about the reference point:²⁰⁹

$$E_X = 2\pi \sum_{\sigma} \iint \frac{\rho_{\sigma}(\mathbf{r})}{s} h_{X\sigma}(\mathbf{r}, s) s^2 ds d\mathbf{r}. \quad (23)$$

The normalization of the spherically averaged hole is

$$N = 2\pi \sum_{\sigma} \iint h_{X\sigma}(\mathbf{r}, s) s^2 ds d\mathbf{r}, \quad (24)$$

which is typically constrained to -1 electron in construction of DFAs.

Figure 5(c,d) shows the negative integrand of the hole normalization (Eqn. 24) for the spherical average the exact and BR exchange holes for two choices of reference point, \mathbf{r} , in H and $\text{H}^{+0.5}$. For the H atom, both the exact and DFA holes are normalized to -1 electron, and the BR hole model is exact by construction, so both curves are overlaid. Conversely, for $\text{H}^{+0.5}$, the normalization integrand for the BR hole is larger in magnitude than that of the exact hole away from the reference point. This occurs because DFA holes, including the BR holes shown in Figure 5, are always assumed to have unit normalization. However, the exact exchange hole in $\text{H}^{+0.5}$ is only normalized to $-1/2$ an electron due to the fractional charge.

Since the exchange energy is determined by the exchange hole via Eqn. 23, having double the normalization from the DFA model hole results in effectively double counting the exchange energy. To illustrate this more clearly, we also plot the exchange-energy density,

$$\varepsilon_X(\mathbf{r}) = 2\pi \rho_{\sigma}(\mathbf{r}) \int h_{X\sigma}(\mathbf{r}, s) s ds, \quad (25)$$

such that the exchange energy is

$$E_X = \sum_{\sigma} \int \varepsilon_X(\mathbf{r}) d\mathbf{r}. \quad (26)$$

Results are shown for both H and $\text{H}^{+0.5}$ in Figure 5(e,f). The BR functional recovers the exact exchange-energy density for H, but gives an exchange-energy density that is too large in magnitude for $\text{H}^{+0.5}$, most noticeably near the nucleus. This results in a large total energy error and explains the behaviour seen for the various DFAs for the H_2^+ PES in Figure 1(a). At equilibrium, the exact exchange hole will have near unit normalization and DFAs provide quite accurate energies. However, as the

dissociation limit is approached, the exact hole normalization will approach $-1/2$ about each nucleus and DFAs will overestimate the magnitude of the exchange contribution to the total energy. An analogous argument can be made for He_2^+ and other stretched odd-electron systems as well.^{4,25-28,210}

Thus, we conclude that delocalization error occurs when the model DFA exchange-correlation hole has a normalization that is too large relative to the exact exchange-correlation hole. This view is consistent with the previous discussion of SIE in Section 2.2, as the Perdew-Zunger SIC can be recast in terms of the exchange hole and serves to reduce the hole normalization.^{63,152}

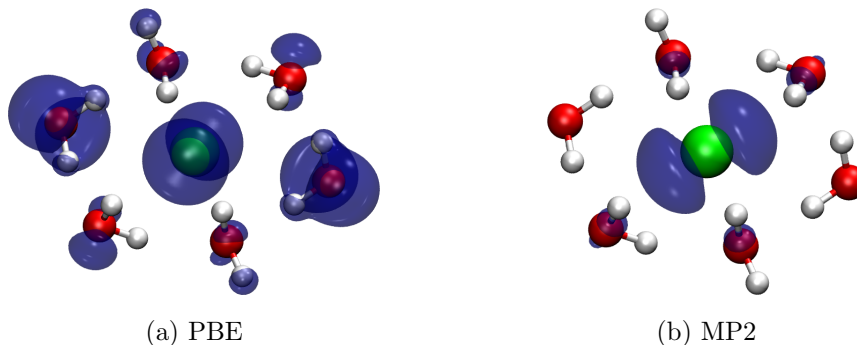
3. ENERGY- VERSUS DENSITY-DRIVEN ERRORS

3.1. Classification of Errors

For all of the examples discussed so far, the electron densities predicted by DFAs are qualitatively correct, and the SIE or delocalization error arises from the DFA energy. However, there are other examples where the DFA electron density itself is qualitatively incorrect, exhibiting an excess delocalization of charge. In such cases, the DFA electron density can be significantly different than what would be expected from correlated wavefunction theory, or even chemical intuition. One such example occurs for the Cl^- anion surrounded by six water molecules to represent the first solvation shell.^{67,202} Here, we would expect the excess charge to remain localized on the Cl atom, but GGA calculations instead delocalize the charge. Figure 6 shows density difference maps for $\text{Cl}^-(\text{H}_2\text{O})_6$ versus $\text{Cl} \cdot (\text{H}_2\text{O})_6$ to visualize the spatial distribution of the excess charge. This illustrates why many-electron SIE is commonly termed delocalization error. Another dramatic example of qualitatively incorrect density distributions from GGAs occurs for solvated-electron model systems.¹⁷² In this case, functionals with delocalization error incorrectly preferred fractional charges and, as the degree of delocalization error increased, unpaired electrons were shared nearly equally between distant moieties.

Sim, Burke, and co-workers coined the terms “energy-driven” and “density-driven” delocalization errors to differentiate between its two manifestations.²¹¹ To quantitatively distinguish between these two classes of errors, we compare the results from the usual self-consistent DFT calculations (SCF) with those from density-corrected DFT,²¹²⁻²¹⁹ in which DFA energies are calculated in a post-SCF fashion using the converged HF orbitals and electron density. The SCF calculations will include both energy- and density-driven errors, while the post-HF calculations will only involve energy-driven errors since the HF densities are free of delocalization error by construction. If using the HF densities significantly reduces or

FIG. 6. Density difference maps for $\text{Cl}^-\cdot(\text{H}_2\text{O})_6$ relative to $\text{Cl}\cdot(\text{H}_2\text{O})_6$ using PBE (0.002 a.u. isosurface) and MP2 (0.006 a.u. isosurface). Calculations were performed with the 6-31+G* basis set using the Gaussian program.¹³⁸



eliminates the delocalization error, then it must be density driven. Conversely, if the SCF and post-HF calculations give similar results, then the delocalization error is energy driven.

To illustrate this analysis, Table 1 compares mean absolute errors (MAEs) and mean errors (MEs) for four benchmark sets obtained using both SCF and post-HF calculations. Results are shown for four functionals with varying extents of exact-exchange mixing: the PBE⁸⁶ GGA (0% HF), the SCAN¹⁰⁶ meta-GGA (0% HF), the PBE0¹²⁵ hybrid (25% HF), and the LC- ω PBE^{133,134} range-separated hybrid (100% HF at large r_{12}). In most cases, the MAEs decrease in the order PBE>SCAN>PBE0>LC- ω PBE, as expected, since SCAN has more inherent self-interaction correction than PBE and increased exact-exchange mixing reduces delocalization errors.

3.2. Energy-Driven Errors

The dissociation limits of the simple ions shown in Figure 1 are all examples of energy-driven errors since the DFA density distributions are essentially correct, with the charge split evenly between the two fragments due to symmetry. To encapsulate these into a simple benchmark, Goerigk, Grimme, and co-workers created the SIE4 \times 4 set,¹¹⁰ which consists of 4 geometries for each of H_2^+ , He_2^+ , $(\text{NH}_3)_2^+$, and $(\text{H}_2\text{O})_2^+$. The high-level reference binding energies, relative to the separated neutral and cation species, were obtained from W4 theory.²²³ The SIE4 \times 4 MAEs in Table 1 decrease only marginally using post-HF densities, confirming energy-driven delocalization error.

Energy-driven errors also occur for systems with extended conjugation, where GGAs overstabilize the delocalized electron density in the π system. In the solid state, this results in overstabilization of graphite relative to diamond,⁵⁵ while in molecular calculations it can affect the energy ordering of C_{20} isomers,⁵⁴ although both of these examples are complicated by the role of London dispersion. A simpler benchmark can be as-

sembled from the work of Woodcock and co-workers on the relative stabilities of cumulenes and poly-yenes.²²¹ For the cumulenes (allene, penta-1,2,3,4-tetraene, and hepta-1,2,3,4,5,6-hexaene), the alternating π -bonds lie in orthogonal planes from each other, inhibiting conjugation. Conversely, for the poly-yenes (propyne, penta-1,3-diyne, and hepta-1,3,5-triyne), the alternating single and triple bonds allow extended conjugation along the full length of each molecule. As a result, GGAs overstabilize the poly-yenes relative to the cumulenes when compared to high-level CCSD(T) reference data.²²¹ The DFA MAEs for this isomerization benchmark in Table 1 again show only small decreases using HF densities, which is indicative of an energy-driven error.

3.3. Density-Driven Errors

Underestimation of barrier heights for simple H-atom transfer reactions, such as the $\text{H} + \text{H}_2$ identity reaction, is a well-known example of delocalization error.³³⁻³⁵ The finding that large amounts of HF exchange mixing, near 50%, yielded improved performance led to development of hybrid functionals targeted for chemical kinetics.^{126,127} Lynch and Truhlar’s BH22 set³³ of (forward and reverse) barrier heights for 22 H-atom transfer reactions is the established benchmark. It has also been known for some time that using HF densities for DFA energy evaluation (i.e. density-corrected DFT) improves accuracy for these barriers.³⁸ A comparison of the SCF and post-HF results for the BH22 set with selected functionals is shown in Table 1 and confirms this to be a case of primarily density-driven delocalization error. This is likely because the GGA and meta-GGA functionals do not sufficiently localize the unpaired electron within the transition-state structure.

The second example of density-driven delocalization error in Table 1 is Bauza’s set of halogen-bonded (XB) complexes,²²² with updated reference data.⁵⁹ Halogen bonding refers to a favourable non-covalent interaction between the negatively charged lone pair of one atom (a pnictogen, chalcogen, or halogen) and the positively

TABLE 1. Mean absolute errors (in kcal/mol) for four delocalization-error benchmarks obtained with four selected DFAs. Results use either self-consistent DFA orbitals (SCF) or converged HF orbitals (HF). Mean errors are shown in parentheses, with negative values indicating overbinding. All calculations were performed using the aug-cc-pVTZ basis set (except for the XB benchmark, which used aug-cc-pVQZ) and Grimme’s D3(BJ) dispersion correction²²⁰ with Gaussian¹³⁸ (psi4¹³⁹ for the SCAN calculations).

Functional	Density	SIE4×4 ¹¹⁰	Isomers ²²¹	BH22 ³³	XB ^{59,222}
PBE	SCF	23.5 (-23.5)	9.8 (-9.8)	10.2 (-10.2)	3.15 (-3.15)
	HF	18.3 (-18.3)	7.3 (-7.3)	3.6 (-3.3)	0.77 (-0.47)
SCAN	SCF	17.9 (-17.9)	8.5 (-8.5)	7.8 (-7.8)	3.41 (-3.39)
	HF	14.1 (-14.1)	6.3 (-6.3)	3.5 (-3.3)	1.42 (-1.15)
PBE0	SCF	14.3 (-14.3)	6.6 (-6.6)	5.1 (-5.1)	2.36 (-2.36)
	HF	11.1 (-11.1)	5.1 (-5.1)	1.7 (-1.1)	0.97 (-0.86)
LC- ω PBE	SCF	9.6 (-9.6)	0.9 (-0.9)	1.5 (-1.0)	0.74 (0.16)
	HF	6.9 (-6.5)	0.3 (-0.2)	2.2 (2.0)	0.64 (0.50)


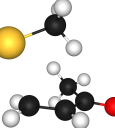
charged σ -hole of a halogen.^{224–227} Analogous interactions can also form involving σ -holes of chalcogen atoms and are termed chalcogen bonding.^{228–230} Both halogen- and chalcogen-bonded complexes tend to exhibit substantial charge transfer and, consequently, are susceptible to density-driven delocalization error.^{59–61} Such errors are magnified in the Bauza set, relative to other halogen-bonding benchmarks,^{231,232} since it contains many complexes involving Cl^- or Br^- anions. As shown in Table 1, use of the HF densities significantly lowers the MAEs obtained with PBE, SCAN, and PBE0, confirming that the errors for this benchmark are density driven.

The BH22 and XB results illustrate the utility of density-corrected DFT,^{212–217} where DFA energies are evaluated in a post-HF fashion in production calculations to eliminate density-driven delocalization error. Density-corrected DFT can also improve performance for many additional benchmarks within the GMTKN55 set²¹⁸ and notably results in excellent performance of the SCAN functional for water.²¹⁹ However, this approach cannot correct for energy-driven errors, nor can it be applied to cases where delocalization error affects predicted structures, unless gradients are implemented for geometry optimization. Additionally, the use of a HF density will not always improve DFA results, particularly for conjugated radicals and some transition-metal compounds, where HF can suffer from severe spin contamination.²³³

3.4. Effects of Delocalization Error on Geometries

In some cases, delocalization error in the energy or density is sufficient to significantly alter molecular geometries. Simple molecular systems where delocalization error alters the potential energy surface and leads to qualitatively incorrect geometries are the $\text{HO}\cdot\text{Cl}^-$ radical anion²¹² and several transition-metal triatomics.²³⁷ Two further examples are shown in Table 2. First is the pre-reaction complex for hydrogen-atom transfer between 1,4-diazabicyclo[2.2.2]octane (DABCO) and the benzy-

TABLE 2. Illustration of density-driven delocalization error on molecular geometry for the benzyloxy-DABCO pre-reaction complex²³⁴ (top) and the enolate intermediate for thio-Michael addition^{235,236} (bottom). Key interatomic distances (R in Å), binding energies relative to the separated molecules (ΔE in kcal/mol), and extent of Mulliken charge transfer (CT in e^-), are shown for three selected DFAs. The DFT calculations were performed using the 6-31+G* basis set and Grimme’s D3(BJ) dispersion correction²²⁰ with the Gaussian program.¹³⁸

	Method	R(C–H)	ΔE	CT
	PBE	1.179	15.2	0.166
	PBE0	1.169	10.1	0.092
	LC- ω PBE	1.129	8.2	0.008
	Method	R(C–S)	ΔE	CT
	PBE	2.187	26.0	0.270
	PBE0	1.966	23.5	0.124
	LC- ω PBE	1.878	21.9	0.060

loxy radical.²³⁴ With GGA functionals, there is considerable charge transfer in this complex. The C–H bond of the benzyloxy radical (nearly perpendicular to the plane of the aromatic ring) becomes stretched and the N–H intermolecular contact compressed, forming a strong C–H–N pseudo hydrogen bond. The second example is the intermediate of the thio-Michael addition reaction of methylthiolate anion to methylvinylketone.^{235,236} While this reaction should proceed via an enolate intermediate, GGA functionals instead optimize the structure to that of a charge-transfer complex without a formal C–S bond. In both cases, the binding energies show the clear trend of decreasing with increased exact-exchange mixing, as is characteristic of delocalization error. Further, the extent

of charge transfer between the interacting fragments also decreases sharply with increased exact-exchange mixing, indicating the error to be density driven.

Even more dramatically, delocalization error can result in changes in protonation state in organic acid-base co-crystals.⁶² Here, geometry relaxation can result in spontaneous proton transfer from the acid to the base, incorrectly predicting neutral co-crystals to exist as organic salts. Examples such as these are particularly challenging tests for methods designed to reduce delocalization error, as they require corrections not just to the density or energy, but also to the energy gradients required for geometry optimization.

4. CONNECTION WITH DISPERSION CORRECTIONS

4.1. DFA Descriptions of Non-Bonded Repulsion

The characteristic signature of delocalization error is improved performance with increased exact-exchange mixing. However, care must be taken to be sure that such situations are in fact caused by SIE and not another systematic error in DFAs. One competing error that can be mistaken for SIE lies in the treatment of non-bonded repulsion, which is controlled by the form of the enhancement factor in most DFA exchange functionals.^{88–96} All GGA functionals share the general form

$$E_X^{\text{GGA}} = \sum_{\sigma} \int \varepsilon_{X,\sigma}^{\text{LSDA}} F(\chi_{\sigma}) d\mathbf{r}, \quad (27)$$

where $\varepsilon_{X,\sigma}^{\text{LSDA}}$ is the LSDA exchange-energy density,

$$\chi_{\sigma} = \frac{|\nabla\rho_{\sigma}|}{\rho_{\sigma}^{4/3}} \quad (28)$$

is the reduced density gradient, and $F(\chi_{\sigma})$ is the enhancement factor. This factor serves to increase the contribution to the exchange energy, beyond that seen with the LSDA, for the large gradient regions that occur in finite molecular systems. The behaviour of various GGAs (and their ensuing hybrids) for dispersion-bound systems can be directly linked to differences in the large-gradient limits of their enhancement factors.^{88–96}

Figure 7(a) shows the exchange enhancement factors of the popular BLYP^{84,121} and PBE⁸⁶ functionals, along with the B86b exchange functional⁸². The optimal behaviour of the enhancement factor for accurate treatment of non-bonded repulsion is to have a large-gradient limit that diverges as $\chi_{\sigma}^{2/5}$, which is obeyed by the PW86⁸³ and B86b⁸² GGAs. The combination of B86b exchange with the PBE correlation functional⁸⁶ (B86bPBE) is our group’s GGA of choice,⁹⁶ although the PBE exchange-correlation functional is vastly more popular. BLYP^{84,121}

gives a large-gradient limit that diverges too quickly, while the PBE⁸⁶ enhancement factor approaches a constant, as they were each designed to obey different constraints. While some researchers prefer “dispersionless” exchange functionals^{96,239} that provide an accurate description of non-bonded repulsion, another strategy in development of DFAs is to attempt to capture “intermediate-range” dispersion through the exchange functional.^{240–242} This is based on the principle that local and semi-local exchange functionals already mimic non-dynamical (or static) correlation and is prevalent in the development of meta-GGAs, such as the Minnesota functionals and SCAN.

To illustrate the connection between enhancement factor and treatment of non-bonded repulsion, Figure 7(b) shows computed potential energy surfaces for the dispersion-bound methane dimer. In the absence of a dispersion correction, BLYP gives a PES that is far too repulsive compared to HF theory, while PBE gives a PES that is too attractive. Thus, the shifts in the PES, relative to the HF reference curve, follow the same ordering as the large-gradient limits of the enhancement factors.

Given the above, it is clear that, in cases where the description of non-bonded repulsion affects the thermochemistry, mixing in some fraction of exact exchange can reduce errors seen with common GGAs, even without significant SIE. As an example, we consider the “DARC” benchmark of 10 Diels-Alder reaction energies from the GMTKN data set.¹¹⁰ In these reactions, there is a loss of π -conjugation in going from the reactants to the products, so delocalization error is expected to play some role.²⁴⁶ However, treatment of non-bonded repulsion is much more important for these addition reactions.²⁴⁶ This can be clearly seen by contrasting the BLYP and PBE results in Table 3. If SIE were the dominant error, then we would expect to see similar performance for both functionals, as for the H_2^+ and He_2^+ PES in Figure 1. However, while both GGAs systematically underbind the addition products, the errors in Table 3 are 3-4 times larger for BLYP than for PBE due to overestimation of non-bonded repulsion by BLYP. Based only on the improvement from BLYP to B3LYP, one might conjecture that SIE is the dominant source of error for these reactions, but this is not reflected in the good performance of dispersion-corrected PBE and SCAN. Finally, the large error seen with LC- ω PBE for the addition reactions is most likely due to poor transferrability of the range-separation parameter between the small-molecule reactants and the much larger products. As highlighted previously, tuning of an optimal range-separation parameter is dependent on system size.^{173,197,200}

4.2. Dispersion Corrections and Performance for Intermolecular Benchmarks

Almost all DFAs (including the LSDA, GGAs, meta-GGAs, global and range-separated hybrids) include only

FIG. 7. Behaviour of the GGA enhancement factor for selected functionals (a). Also shown are potential energy curve for the methane dimer (b) with selected methods computed using FHIaims¹⁷⁵ with tight settings for the DFT calculations and Gaussian¹³⁸ with the aug-cc-pVTZ basis set for HF.¹³⁸ Panel (c) illustrates the effect of the XDM dispersion correction; for comparison, the reference CCSD(T) binding energy of the methane dimer is 0.53 kcal/mol.²³⁸

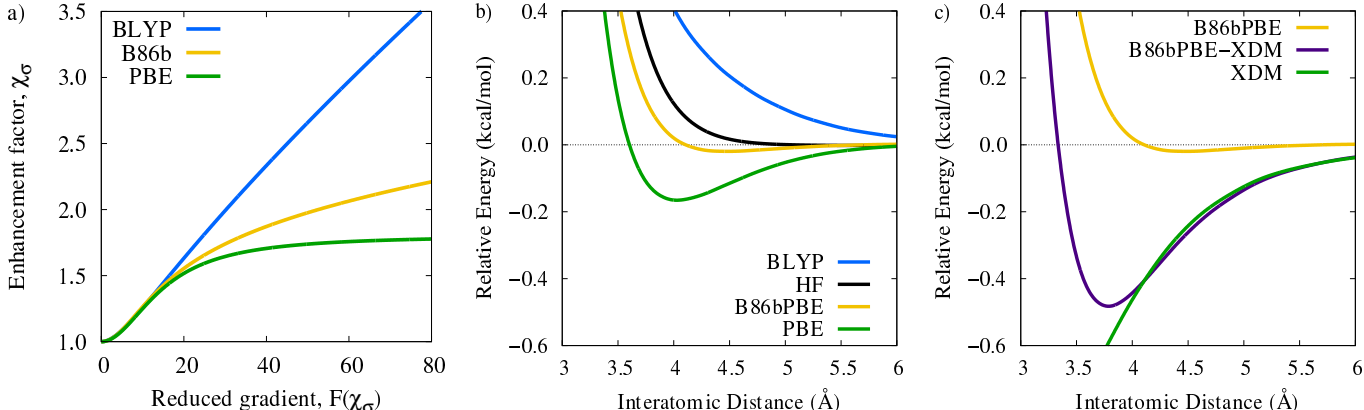


TABLE 3. Mean absolute errors (in kcal/mol) for several literature benchmarks obtained using selected functionals with and without a dispersion correction. Mean errors are given in parentheses, with negative values indicating overbinding. All calculations were performed using the aug-cc-pVTZ basis set (except for the XB benchmark, which used aug-cc-pVQZ) with Gaussian 16¹³⁸ (psi4¹³⁹ for the SCAN calculations).

Functional	Dispersion	DARC ¹¹⁰		S66 ^{243,244}		HB ²⁴⁵		XB ^{59,222}	
BLYP	none	23.1	(23.1)	4.07	(4.07)	6.32	(6.32)	1.85	(1.02)
	D3(BJ)	13.7	(13.7)	0.23	(-0.09)	2.10	(-2.10)	1.72	(-1.10)
B3LYP	none	15.3	(15.3)	3.16	(3.16)	3.79	(3.79)	1.32	(0.64)
	D3(BJ)	7.5	(7.5)	0.34	(-0.33)	3.14	(-3.14)	1.25	(-1.14)
PBE	none	6.8	(6.6)	2.08	(2.08)	2.05	(-2.05)	2.56	(-2.22)
	D3(BJ)	3.2	(2.0)	0.32	(-0.27)	6.65	(-6.65)	3.15	(-3.15)
SCAN	none	2.4	(0.1)	0.76	(0.52)	5.74	(-5.74)	3.25	(-3.17)
	D3(BJ)	1.9	(-1.2)	0.45	(-0.43)	7.27	(-7.27)	3.41	(-3.39)
PBE0	none	3.0	(0.2)	1.98	(1.97)	0.61	(0.01)	1.70	(-1.48)
	D3(BJ)	4.0	(-4.0)	0.33	(-0.31)	4.05	(-4.05)	2.36	(-2.36)
LC- ω PBE	none	6.2	(-6.2)	2.42	(2.42)	4.71	(4.71)	1.42	(1.19)
	D3(BJ)	10.8	(-10.8)	0.26	(-0.10)	0.45	(0.44)	0.74	(0.16)

local or semi-local ingredients to model electron correlation and, hence, do not include the requisite physics to model London dispersion.⁹⁶ While some DFAs mimic dispersion-like binding at short intermolecular distances, this is a spurious attraction arising from the exchange functional rather than a physically meaningful representation of dispersion, and the correct $-C_6/R^6$ asymptotic behaviour is lacking. Thus, most DFAs should be paired with a dispersion correction. Dispersion can be effectively modeled by an additive post-SCF correction to the DFT energy, as is done for the Grimme,^{220,247} Tkatchenko-Scheffler,²⁴⁸ many-body,^{249,250} or our own exchange-hole dipole moment (XDM),²⁵¹⁻²⁵⁴ dispersion corrections. Alternatively, explicitly non-local DFA models of dispersion can be used, as in the vdW-DF2^{255,256} and rVV10²⁵⁷ functionals. The importance of a dispersion correction is emphasized in Figure 7(c) for the methane dimer and by the MAEs in Table 3 for the S66 set^{243,244} of van der Waals complexes. We selected

D3(BJ) for all examples in Tables 1-3 since it is the most popular option with users of molecular DFT.

In all cases, the dispersion correction adds necessary physics missing from the underlying DFA and should not be viewed as optional. However, there have been reports of dispersion corrections worsening performance for certain benchmarks,^{222,232} so dispersion is unfortunately not always included in production DFA calculations on intermolecular complexes or on large molecular systems, where it can significantly impact the chemistry. The reason for the apparent contradiction of dispersion corrections worsening results is due to the interplay between dispersion with both the exchange enhancement factor and delocalization error. The differing behaviour of base DFAs for non-bonded repulsion explains why additive dispersion corrections should be parameterized individually for use with a given functional to benefit from error cancellation.¹¹⁰

Dispersion is an attractive interaction and will always

stabilize vdW dimers, supermolecular complexes, and molecular crystals, to list only a few examples. However, if these systems are already overstabilized because of delocalization error or an exchange functional that already mimics intermediate-range dispersion-like binding, then adding a dispersion correction will increase the extent of the error. Examples are shown in Table 3 for the Water20 hydrogen-bonding benchmark²⁴⁵ (HB) and Bauza’s halogen-bonding benchmark^{59,222} (XB).

The Water20 results can be explained by the exchange enhancement factor. BLYP and B3LYP are overly repulsive, so adding dispersion decreases their MAEs; this actually changes the sign of the overall error and leads to overbinding due to delocalization error since GGAs are known to excessively polarize co-operative hydrogen-bonding networks.^{95,96,164,219,258} On the other hand, PBE and SCAN are already overly attractive without dispersion. For both these functionals, and for PBE0, which coincidentally gives quite good performance for H-bonds in the absence of dispersion, adding a dispersion correction increases the MAEs. Finally, LC- ω PBE provides a reasonable description of non-bonded repulsion and minimal delocalization error, so adding dispersion once again improves the results, providing the lowest MAE.

In Section 3.3, we saw that there is significant density-driven delocalization error for the halogen-bonding benchmark, where GGAs predict excessive charge transfer. Thus, delocalization error leads to overbinding of the halogen-bonded complexes with PBE, SCAN, and PBE0, so adding a dispersion correction once again increases the error. As LC- ω PBE has minimal delocalization error, adding dispersion improves performance. Dispersion also slightly improves performance for BLYP and B3LYP, as the errors in these functionals for non-bonded repulsion partially offset the delocalization error.

In summary, when selecting a DFA, it is important to choose based on inclusion of appropriate physics and the interested reader is directed to our perspective on requirements for an accurate dispersion-corrected DFA.⁹⁶ Care must be taken not to simply choose a DFA based on good performance for one or two literature benchmarks when this may be entirely due to fortuitous error cancellation.

5. OUTLOOK: METHOD CHOICES AND ONGOING DFA DEVELOPMENT

Now that we have reviewed the origins of delocalization error, its decomposition into energy-driven and density-driven contributions, and its interplay with dispersion corrections, what strategies should be taken to reduce it in production calculations? The most straightforward method to reduce delocalization error is use of hybrid or range-separated hybrid functionals. The improvement in properties going from GGAs to conventional hybrids with 20-25% exact exchange, to hybrids with near 50% exact exchange, is the typical indicator of systems with significant SIE. The drawback here is that one must rec-

ognize when delocalization error is important in a system and choose a hybrid functional accordingly. As stated previously, a dispersion correction should be paired with all such DFAs to model important, otherwise missing, physics.

Beyond global hybrids, range-separated hybrids with full long-range exact exchange provided the first functionals that can be free of N -electron SIE and these are the most accurate DFAs in common use today.^{110,114} The limitation of range-separated functionals is that the optimal range separation is highly system dependent,^{173,197–200} so a single parameterization cannot be used across main-group and transition-metal chemistry, or for a mix of large and small molecules, with the expectation of full transferability. Additionally, any amount of long-range exact exchange in global or range-separated hybrids is not recommended for bulk metals or small-gap semi-conductors.

Even more accurate than range-separated hybrids for finite molecules are double hybrids,^{259–265} which include correlation-energy terms based on MP2 theory. The use of non-local MP2 correlation is conducive to higher exact-exchange mixing fractions and, hence, almost entirely eliminates delocalization error. However, due to the inclusion of MP2 correlation, the energy depends on the unoccupied (virtual) KS orbitals and, consequently, double hybrids are not true density functionals. By a strict definition, a DFA energy should depend on properties of the electron density and occupied orbitals only.⁷⁴ Inclusion of the MP2 correlation term leads to two unfavourable consequences. The first is that it degrades the scaling with system size, inhibiting application of double hybrids to very large systems. The second is that double hybrids can perform poorly in cases where MP2 has well-known failures, including susceptibility to spin contamination²³³ (albeit to a much lesser extent than MP2 itself for organic radicals²⁶⁶) and divergence of the energy for small HOMO-LUMO gaps, as occur for highly stretched bonds.^{267,268} Similarly to global or range-separated hybrids, double hybrids are not recommended for bulk solids and, due to the zero band gap, will diverge for metals. The solid-state analogue of double-hybrid functionals is the random-phase approximation (RPA). RPA-based methods^{269–274} have also shown great promise at reducing delocalization error but, again, their high computational cost makes them less practical for large systems or crystals with large unit cells.

For very large systems, or for solid-state calculations using plane-wave basis sets, even hybrid calculations can be prohibitively expensive, so methods to reduce delocalization error while maintaining GGA cost are needed. These include Fermi-Löwdin-orbital SIC^{140,154,157–159} and LOSC,^{202–204} as discussed in Sections 2.3 and 2.4, respectively. Another possibility is “rung 3.5” functionals,^{275–278} so named because they are intermediate between meta-GGAs and hybrids in terms of accuracy. These functionals quantify delocalization by projecting the density matrix formed from

the KS orbitals on a set Gaussian-type orbital of atomic size. This projection is done for every reference point on the integration grid and incorporated into the exchange-correlation energy expression, serving as a correction for fractional charges.

Some alternative low-cost strategies are targeted to more niche applications than to computations across the full breadth of chemistry. One approach targeting transition metals is the addition of a Hubbard U term^{279–286} to the exchange-correlation energy that serves to shift the orbital energies of the d or f electrons, preventing hybridization. While GGA+ U methods are capable of high accuracy for metal-containing systems, this approach is not transferable to other manifestations of delocalization error. Another approach is the use of optimized effective potential methods, such as TB-mBJ,^{287,288} where one models the exchange-correlation potential rather than the exchange-correlation energy. Such methods can significantly improve predictions of the band gap.^{289–294} However, as they only provide single-particle (orbital) energies, and not a well-defined total energy expression, they are not applicable to thermochemistry.

The final class of functionals we will discuss is those that depend on the exact exchange-energy density (ϵ_X^{HF}).^{4,14,15,179,295–299} These include local hybrid functionals,^{180,300–306} in which the exact-exchange mixing is position dependent:

$$E_X^{\text{LH}} = \int a_X(\mathbf{r})\epsilon_X^{\text{HF}} d\mathbf{r} + \int [1 - a_X(\mathbf{r})]\epsilon_X^{\text{DFA}} d\mathbf{r}, \quad (29)$$

where the mixing, $a_X(\mathbf{r})$, can be a functional of the density and its derivatives. Alternatively, Becke^{4,14,179} has developed real-space density-functional models of electron correlation that, unlike semi-local DFA correlation functionals, are designed to pair with full exact exchange. Here, the energy is a sum of the HF exchange and the real-space correlation energy, where the latter involves a sum over dynamical,³⁰⁷ non-dynamical,^{4,179} and perhaps strong,¹⁴ correlation-energy terms. The correlation energies themselves are complicated functionals of the density, its derivatives, and the exact exchange-energy density. While this class of functionals is perhaps the most promising avenue for consistent elimination of delocalization error,^{32,308,309} their form has typically made self-consistent implementation impractical, which has all but prohibited their use in production calculations.

The first functional employing the exact exchange-energy density to have a practical and open-source self-consistent implementation is DeepMind 2021 (DM21), which is a local, range-separated hybrid functional developed by Kirkpatrick *et al.*³¹⁰ By training a deep neural network on small main-group molecules (containing elements from H-Kr) in addition to fractional-charge and fractional-spin constraints for H-Ar atoms, this functional attempts to correct delocalization and multi-reference errors for a variety of chemical systems. The input features to the neural network include the spin-

indexed density ρ , $|\nabla\rho|$, the kinetic energy density τ , and the range-separated exact exchange energy densities ϵ_X^{HF} , and $\epsilon_X^{\omega\text{HF}}$. As one would expect, DM21 outperforms existing global hybrids in challenging systems due to its dynamic mixing of exact exchange.³¹¹ While this functional has shown exceptional performance for a number of molecular benchmarks,^{110,310–313} it has not been trained for transition-metal chemistry, extended systems, liquids, or solids.³¹⁴

All of the areas discussed here are highly fruitful ground for ongoing functional development and the above is by no means a comprehensive list of DFA-based approaches to reduce delocalization error. However, despite decades of research effort, a DFA that is free of delocalization error, has consistently high accuracy, and is generally applicable to molecular and solid-state chemistry is still tantalizingly out of reach. In our opinion, resolving delocalization error remains the greatest outstanding challenge in DFT today.

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CONFLICT OF INTEREST

The authors have declared no conflicts of interest for this article.

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